



西昌学院
XICHANG UNIVERSITY

西昌学院相关情况汇报

主讲人：焦 钰

E-mail: jiaoyu@xcc.edu.cn

2026/07/02

目录

CONTENT

I 整体办学情况

II 材料与能源学院

III 可能有的研究合作点





“川西明珠，航天之城。
襟山带湖，星日同辉”
(卫星、阳光、月亮)

是国内少有的“战略矿产 + 零碳能源”
双富集工业基地

资源禀赋：

- 位于川滇巨型成矿带，探明矿种 103 种、具备规模储量 60 种，特大型 / 大型矿床 30 处；含 29 种国家战略性矿产，13 种矿产保有储量四川省第一，矿石普遍品位高、埋藏浅、共伴生多、综合利用价值极高
- 全州清洁能源技术可开发总量 2.25 亿千瓦，水电调节能力全国顶尖，形成水风光互补一体化，解决高载能工业低碳、低成本用电痛点



肇始于 1939 年创建的国立西康技艺专科学校，开启了攀西民族地区高等教育的序幕。

抗战时期中国高等教育迁徙示意图



北洋工学院



西安临时大学



国立西北联合大学



国立西康技艺专科学校

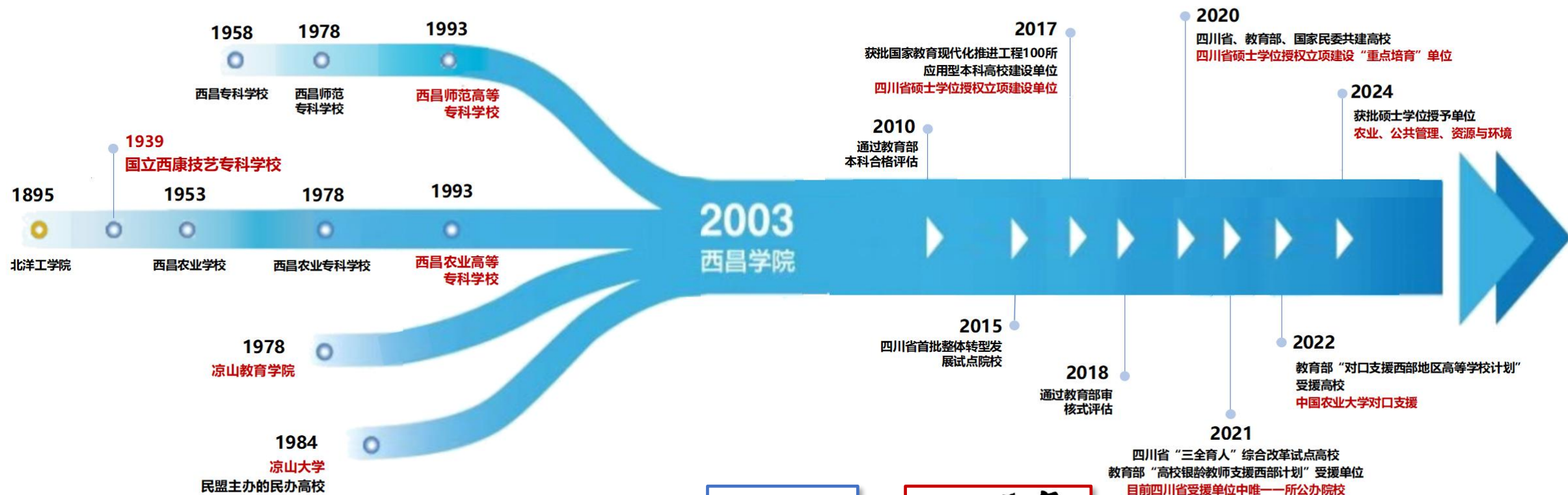


國立西康技藝專科學校遺址



艰苦奋斗
兴学强国

办学初心始终如磐、文化传承使命在肩



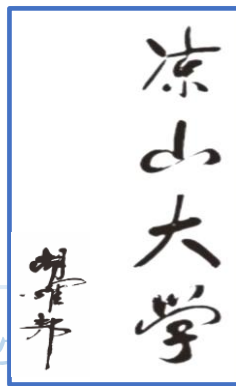
西昌学院
XICHANG UNIVERSITY



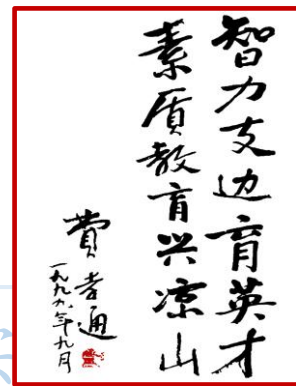
两度毅然迁址、服务大局



明德



乐学



实践





材料与能源学院介绍



稀土、钒钛新材料研究平台 邛海实验室



州校共建攀西战略资源 创新开发重点实验室



省中试平台





实验设备



JT-150注塑机



TSJ-35B挤出机



HT-25T平板硫化机



IEST302E流变仪



XSK-160双辊塑炼机



盐雾试验机



JH-XBL-A冲击试验机



紫外光耐候试验机



实验设备

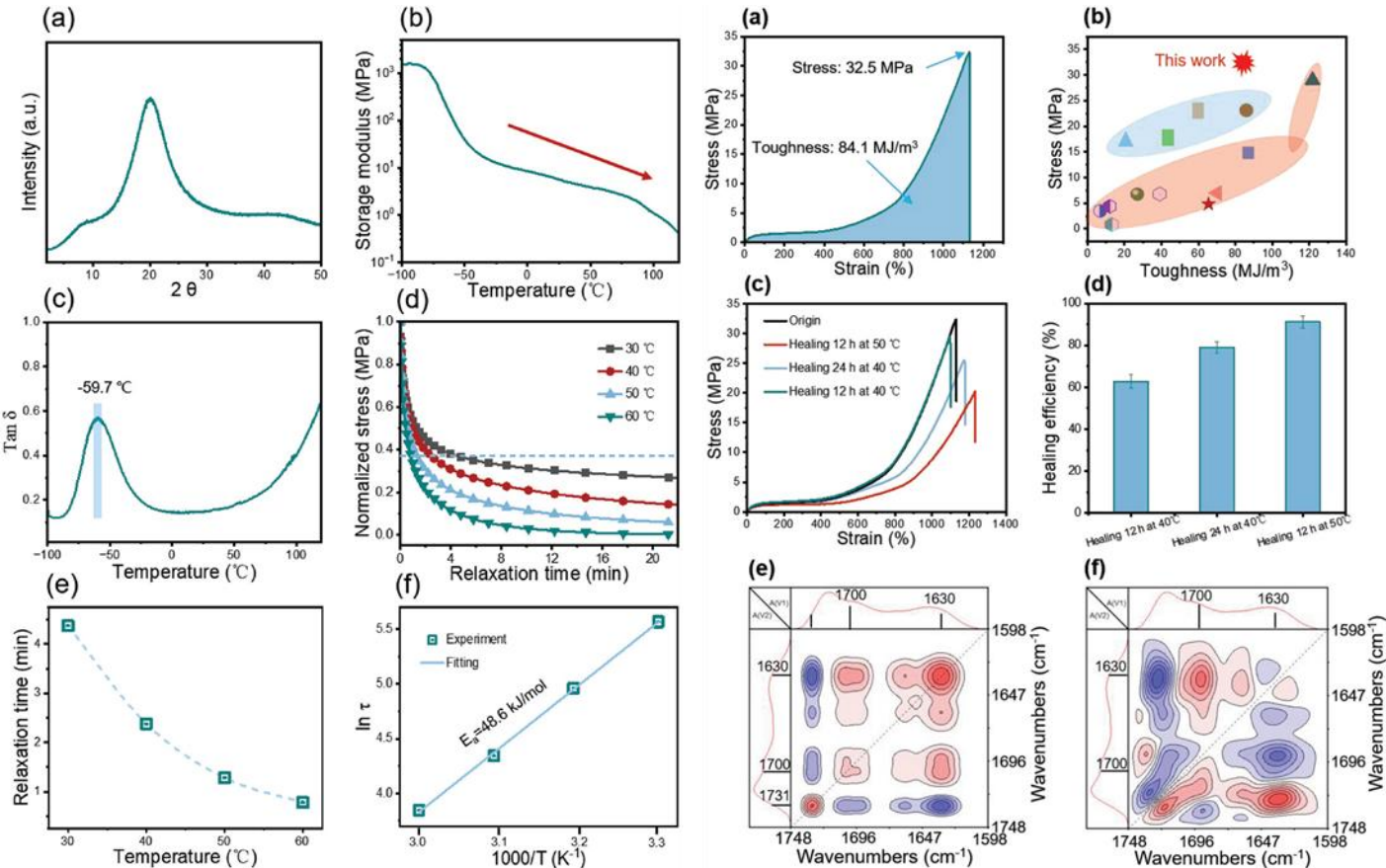


实验室磁重浮选矿设备

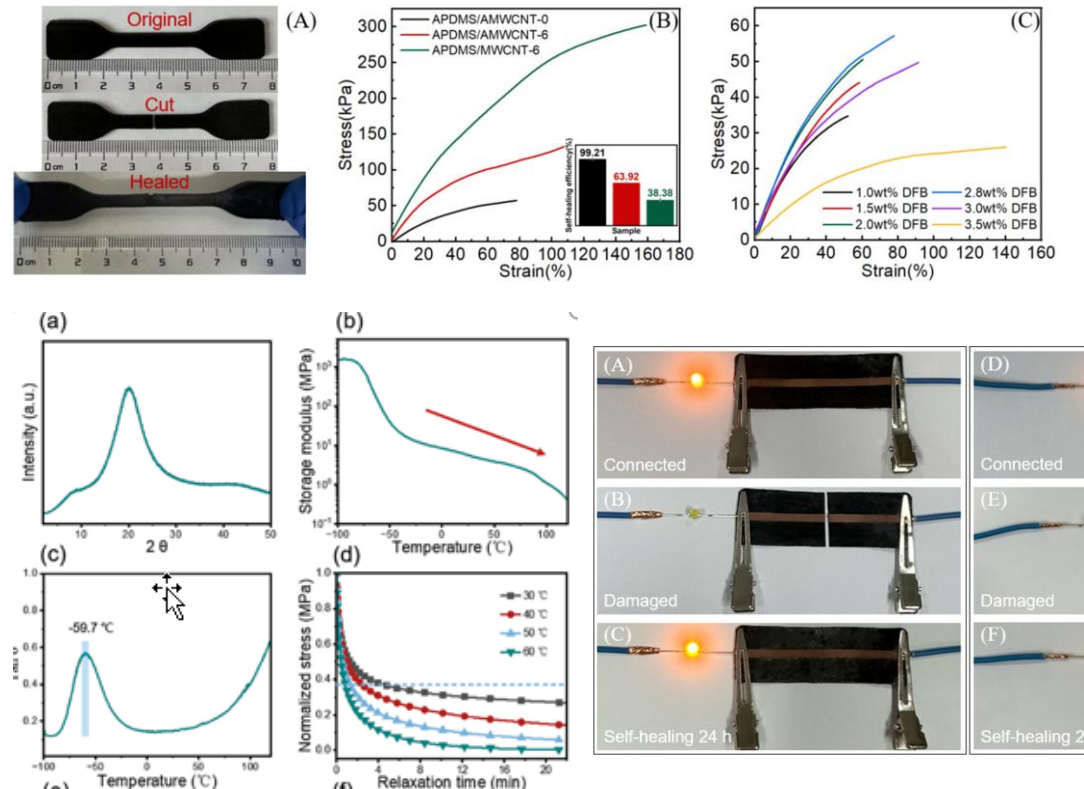
矿物选冶中试线



分层氢键结构提升机械强度与自修复效率

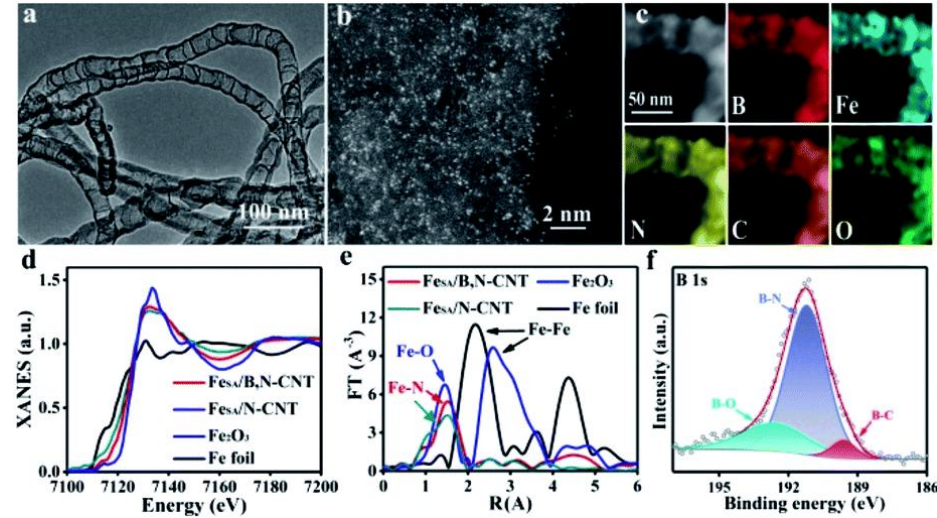
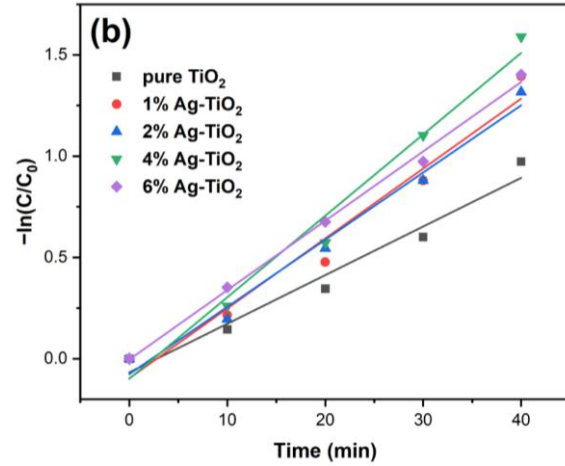
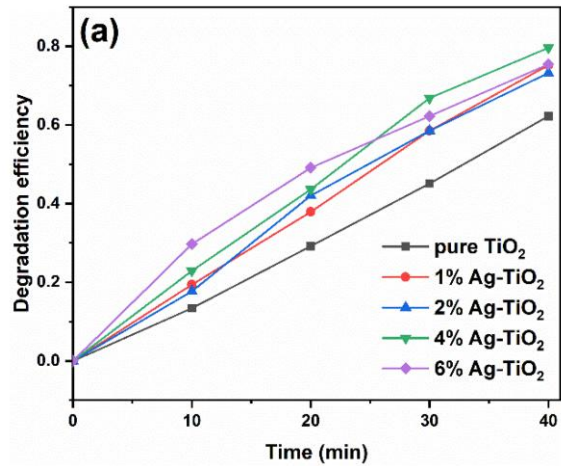
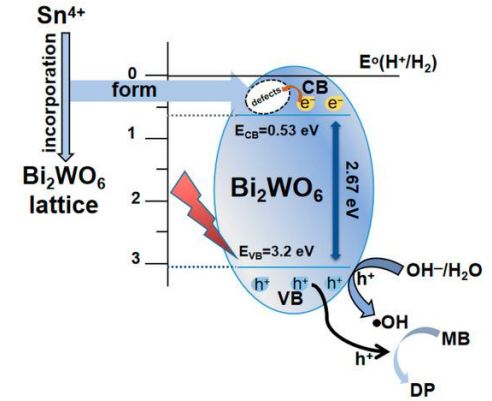
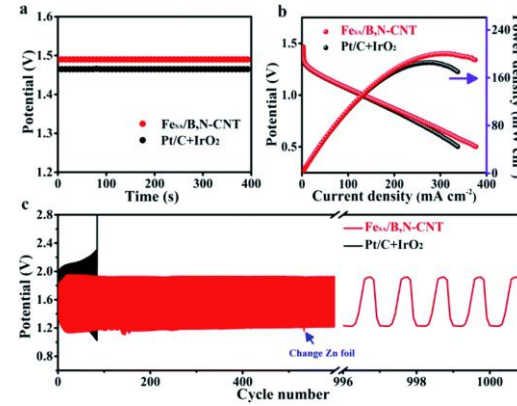
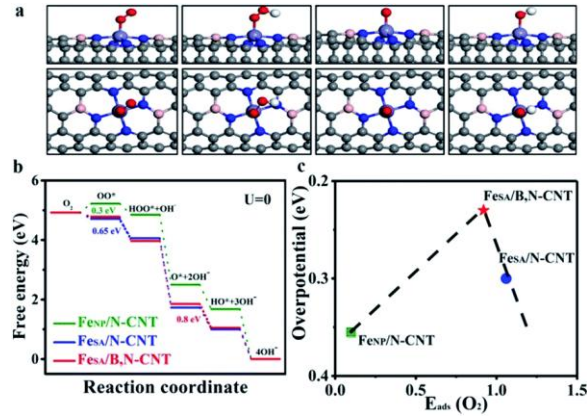
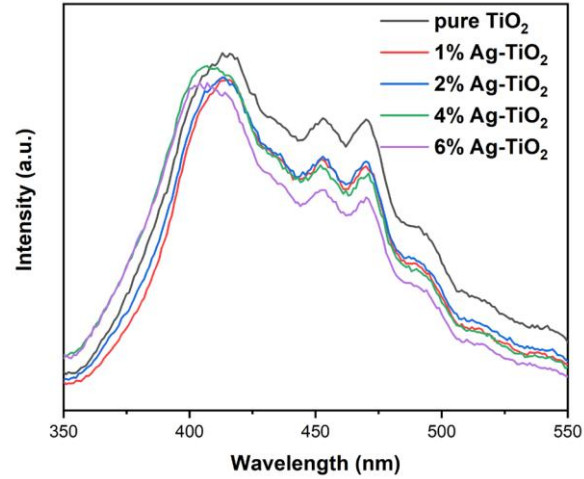


亚胺辅助自修复APDMS/AMWCNT弹性体



分层氢键作为动态交联点，能够有效耗散大量能量。拉伸时，分层氢键逐步断裂，从而实现PU弹性体的增强与增韧

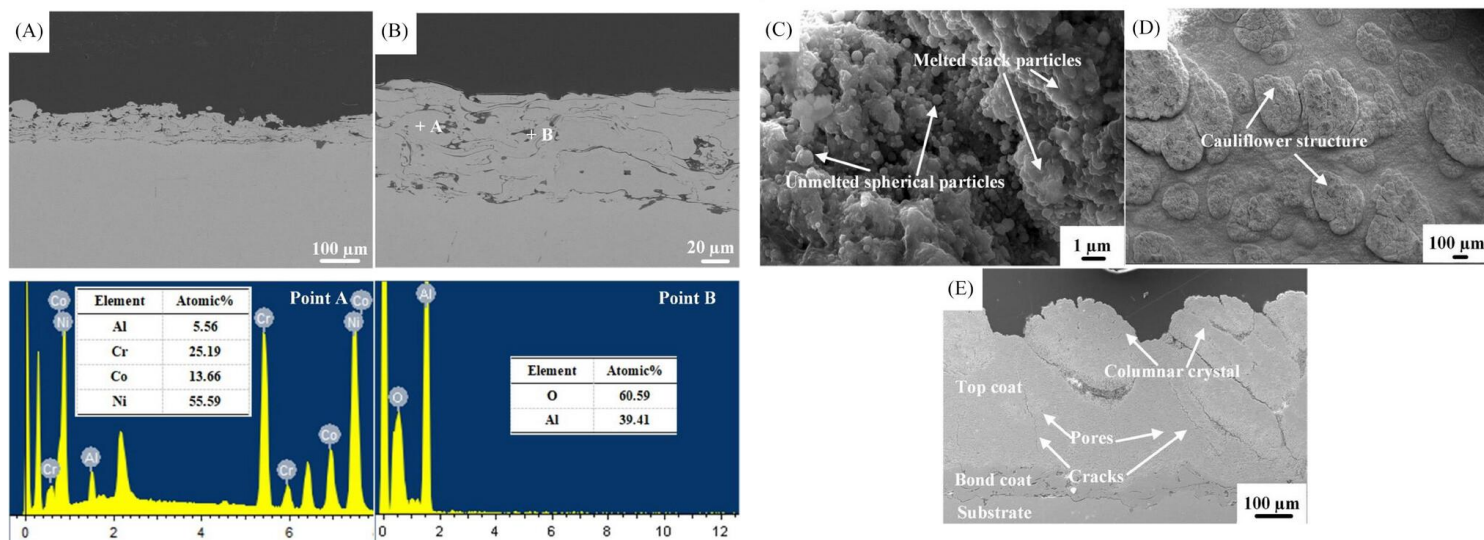
印刷电路应用时可自修复实现电路导通



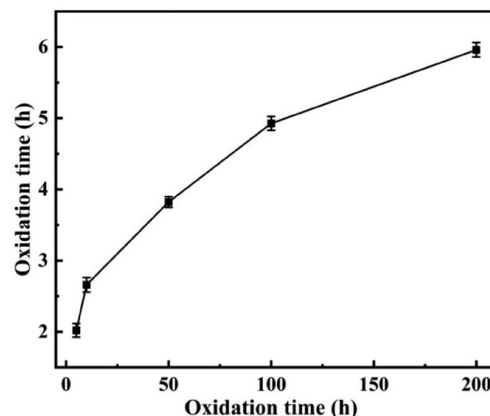
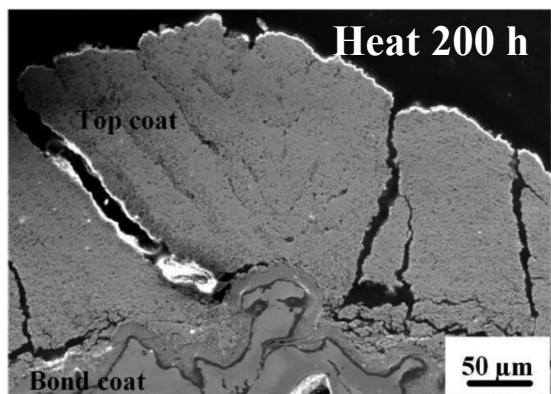
通过掺杂等方式提高材料光催化活性，运用于矿物选冶污水处理



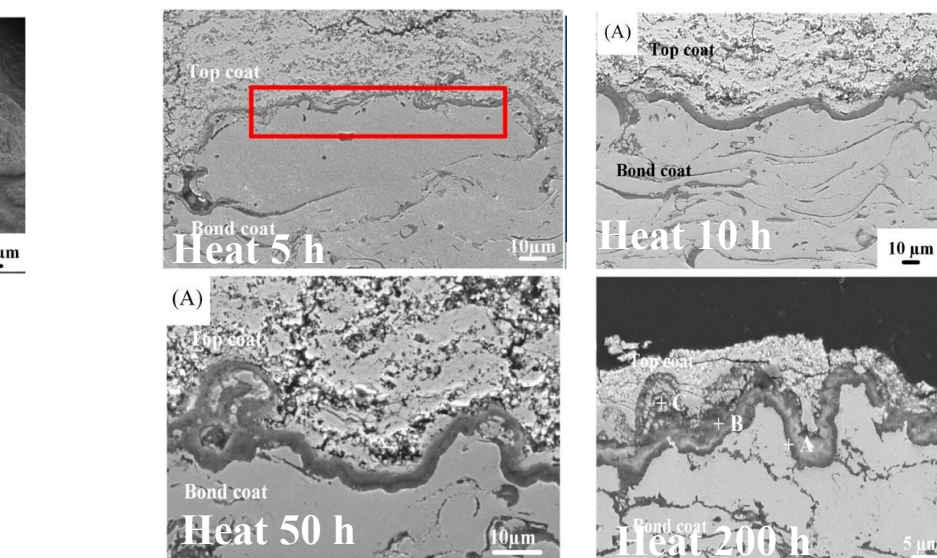
热障涂层



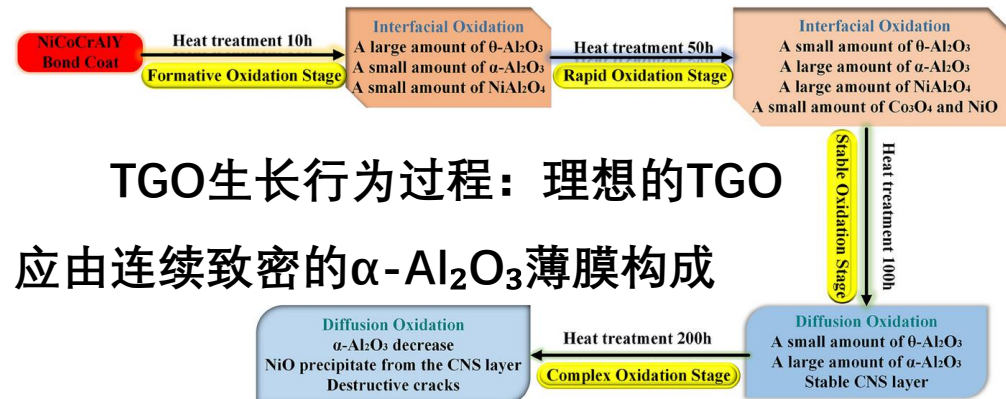
高温软化合金粉末并撞击基底形成层状涂层



TC出现较宽的裂纹, 工作寿命殆尽 TGO的厚度随氧化时间延长而增加



尖晶石结构扩大,CNS组织迅速生长, 形成TGO层中的薄弱点, 进而影响涂层的热循环寿命





项目批准号: 52102307
申请代码: E0208
归口管理部门:
依托单位代码: 51501308A1 689-1930



国家自然科学基金 资助项目计划书 (包干制项目)

资助类别: 青年科学基金项目

亚类说明:

附注说明:

项目名称: 高能量密度高安全性锂金属电池的制备及性能调控

资助经费: 30万元

执行年限: 2022.01-2024.12

负责人: 焦钰

通讯地址: 四川省凉山州西昌市西昌学院

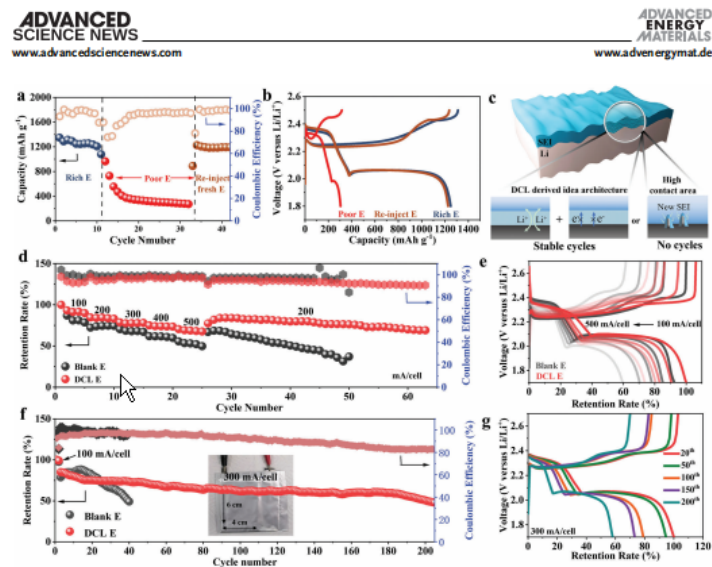


Figure 4. Electrochemical performances of high-energy Li-S pouch cells. (a) Typical cycle performance of Li-S pouch cells not exceeding 10 cycles while the cycle stability is improved again once the fresh electrolyte is injected. The electrolyte during discharge/charge voltage plateau in panel (a). (b) Discharge/charge voltage plateau of DCL electrolyte. (c) Schematic illustration of the formed SEI derived by DCL or blank electrolyte. (d) Rate performance and (e) corresponding voltage plateau of DCL electrolyte at various current density. (f) Cycle performance at a current density of 300 mA per cell and (g) corresponding voltage plateau of DCL electrolyte.

demonstrate that the effectiveness of our strategy using DCL electrolyte (Tables S1 and S2, Supporting Information). The strategies reported in literature are divided into cathode modification and configuration optimization. The cathode modification concerns more about inhibiting the shuttle effect of LiPS, while the configuration optimization concerns more about achieving a stronger SEI structure to passivate the lithium anode. As shown in Table S1 (Supporting Information), the reported Li-S pouch cell can hardly achieve ≈ 20 cycles only by suppressing the shuttle effect of LiPS. By contrast, inhibiting the growth of Li dendrites achieves obvious promotion but so far still not able to maintain >100 cycles (Table S2, Supporting Information). These results further verify the advantage of DCL electrolyte for the commercial use in Li-S pouch cells.

At last, the safety of assembled Li-S pouch cells is demonstrated via nailing (Figure 4) and Figure S21, Supporting Information) and shorting (Figure S22, Supporting Information). We found that the safety was comprehensively superior than that of the lithium-ion batteries. No leakage or explosion occurred whether using DCL electrolyte or blank electrolyte, and the local heat of the nail was slightly lower using DCL electrolyte. For

the short-circuit tests, the maximum temperature of both electrolytes only rises to $\approx 50^\circ\text{C}$, which is critical for the operation of Li-S cells in extreme environments, again illustrating the great potential of DCL electrolyte to promote the commercialization of Li-S cells.

3. Conclusions

Expanding the lithium-sulfur cells from the laboratory coin-scale to pouch-format is a necessary step for the commercialization. In this work, we demonstrate that promoting the stability of lithium metal anode is most urgent rather than inhibiting the polysulfide shuttle effect for improving the performances of Li-S pouch cell. Specifically, KHDF salt is used to regulate the preferred components of SEI, in which KHDF salt decomposes prior to solvent. The inner layer of SEI is successfully modulated to KF and LiF enriched structure, which can effectively alleviate the decomposition of the electrolyte and improve the cycling stability of Li-S pouch cells. This strategy enables the realization of unprecedented long-term cycles (>200 cycles) and high rate (500 mA per cell) characteristics at the format of Li-S pouch cell with sulfur

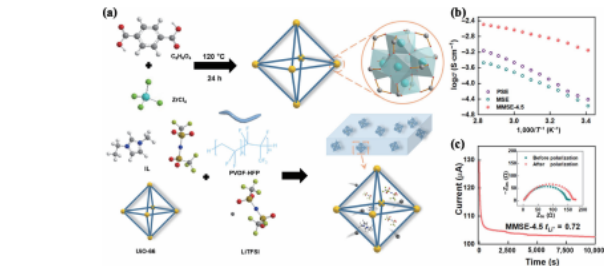


Figure 8. (a) Synthetic routes of UO-66 and MMSE. (b) The ionic conductivity and (c) lithium transfer number of the electrolytes. Reproduced with permission from Ref. [8].

Table 1. Summary of novel electrolytes for Li-S batteries working at low temperature

Recipe	Lowest temperature capacity/rate/loading	Temperature/cycles/final capacity/rate	References
1 M LiFSI in DEE	$-60^\circ\text{C}/474 \text{ mAh g}^{-1}/0.1 \text{ C}/35 \text{ mAh cm}^{-2}$	$-60^\circ\text{C}/50/\text{NA}/0.2 \text{ C}$	[44]
0.7 M LiTFA + 0.2 M LiNO ₃ in (DOL-DME, 85:15 by volume)	$-40^\circ\text{C}/250 \text{ mAh g}^{-1}/0.05 \text{ C}/385 \text{ mg cm}^{-2}$	NA	[51]
1 M LiFSI in (DOL-TTE, 1:4.5 by volume) with 10% FEC	$-30^\circ\text{C}/273.8 \text{ mAh g}^{-1}/0.1 \text{ A g}^{-1}/3.2 \text{ mg cm}^{-2}$	$-30^\circ\text{C}/50/\text{NA}/0.1 \text{ A g}^{-1}$	[63]
LiFSI + DME + FB (1:1.2:3 by molar)	$-20^\circ\text{C}/1,250 \text{ mAh g}^{-1}/0.1 \text{ C}/5 \text{ mg cm}^{-2}$	$-60^\circ\text{C}/300/626 \text{ mAh g}^{-1}/0.5 \text{ C}$	[65]
1 M LiFSI in (MP-FEC, 9:1 by volume)	$-40^\circ\text{C}/450 \text{ mAh g}^{-1}/0.1 \text{ A g}^{-1}/1.5 \text{ mg cm}^{-2}$	$-40^\circ\text{C}/100/387 \text{ mAh g}^{-1}/0.2 \text{ A g}^{-1}$	[69]
7 mg mL ⁻¹ BNNS in electrolyte (1 M LiTFSI in 1:1 v/v DOL/DME with 2 wt % LiNO ₃)	$-20^\circ\text{C}/530 \text{ mAh g}^{-1}/0.1 \text{ C}/15 \text{ mg cm}^{-2}$	NA	[77]

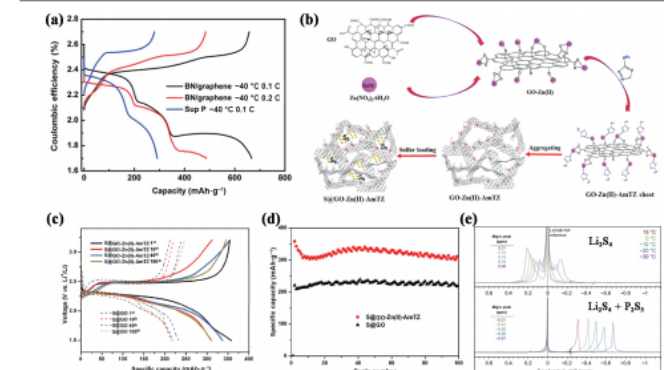


Figure 9. (a) First charge and discharge voltage profiles of BN/graphene and Super P at -40°C . Reproduced with permission from Ref. [94]. (b) Illustration of the preparation of GO-Zn(II)-AmTZ and the SpGO-Zn(II)-AmTZ composite. (c) The first charge and discharge voltage profiles of SpGO-Zn(II)-AmTZ cathodes from different cycles at -20°C . Reproduced with permission from Ref. [94]. (d) The cycling performances at -20°C . Reproduced with permission from Ref. [94]. (e) The variable-temperature ^1D NMR of $0.1 \text{ M Li}_2\text{S}$ and $0.053 \text{ M P}_2\text{S}_5 + 0.1 \text{ M Li}_2\text{S}$. Reproduced with permission from Ref. [94].

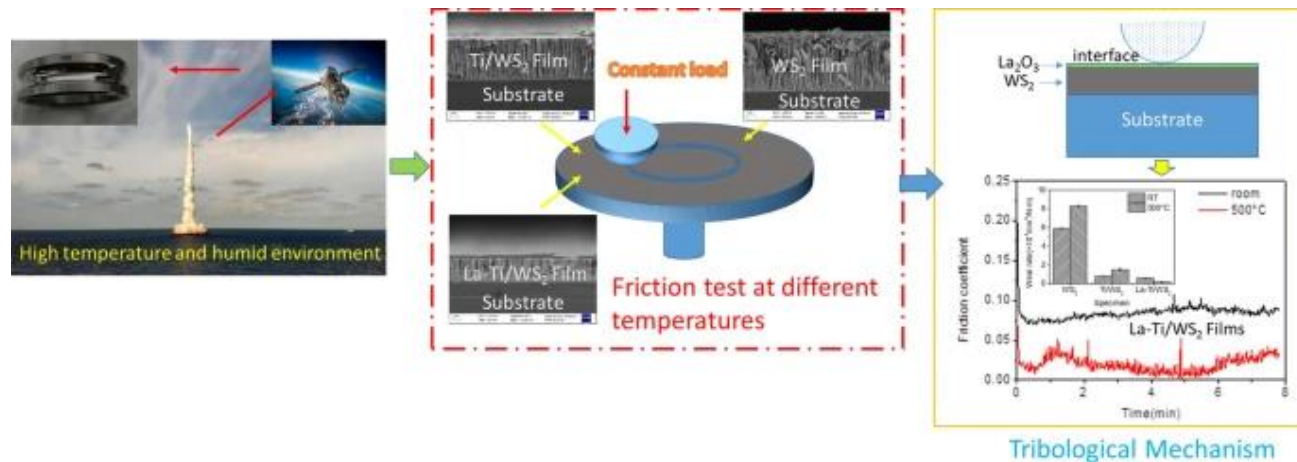
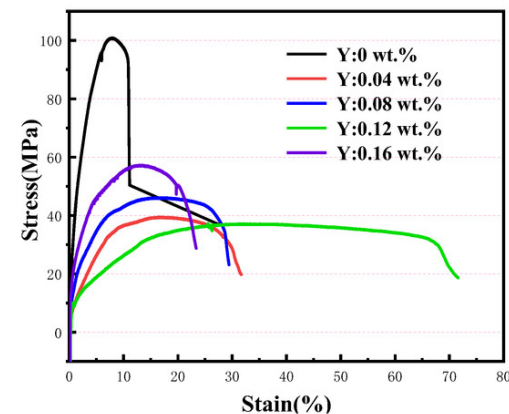
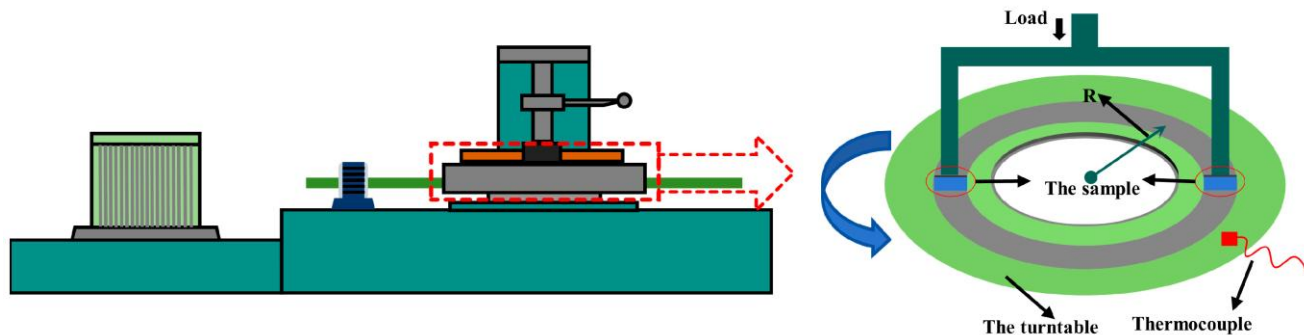
Metal-organic framework (MOF) has been widely used in the electrochemical related field [89–91], especially in Li-S battery due to its porous structure and abundant polar sites [92, 93]. Burns et al. incorporated lithium thiophosphate in Zr-based MOF, which served as LiPS anchoring sites to disrupt the clustering of LiPSs and prevent the film-like Li₂S deposition at low temperature [94].

www.theNanoResearch.com | www.Springer.com/journal/112274 | Nano Research

在自科基金资助下，锂电池研究上有所进展



可能有的研究合作点



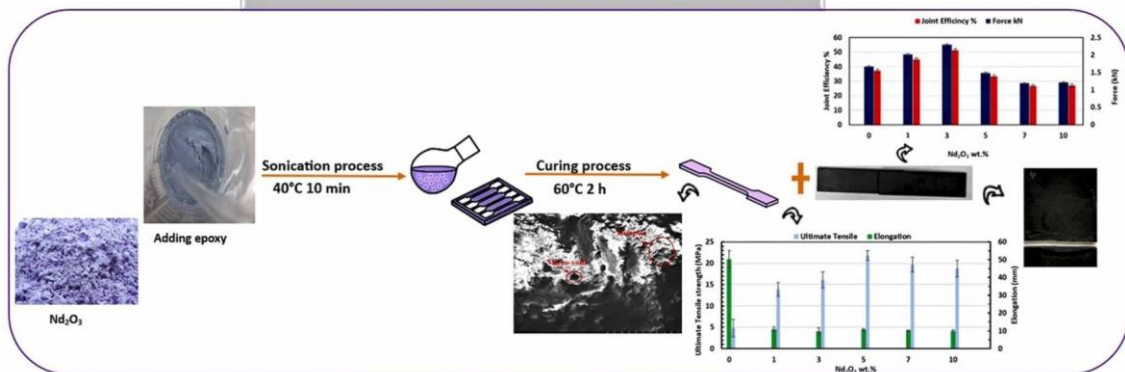
是否可以共同选别可用于中微子实验的PMMA、液闪材料？

是否可以针对实验室所需材料对市场产品提纯或改性？



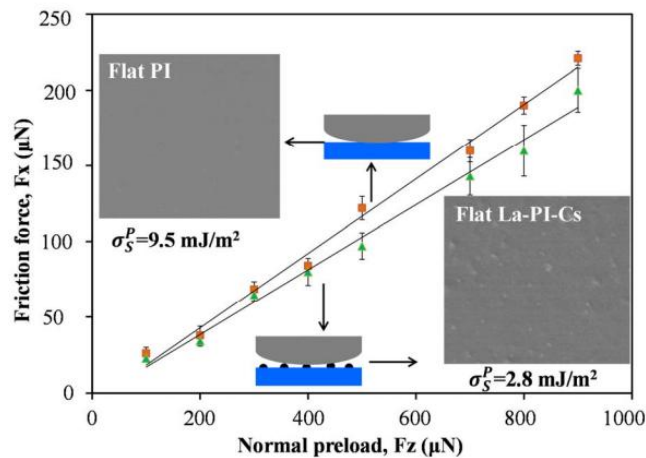
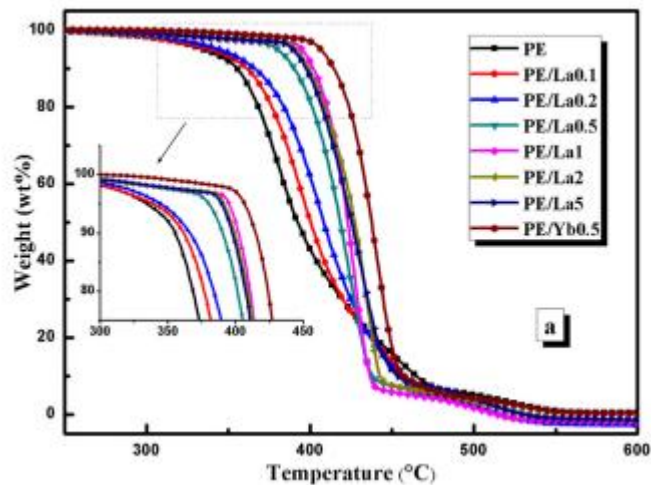
可能有的研究合作点

Develop Adhesive Material for Joining CFPEEK



- Tensile stress Properties
- Adhesive joint efficiency
- Lap shear strength properties
- Microstructure characterizations

是否可以共同开发本地战
略资源的更广泛应用场景？





团队成员



焦钰 教授



郑春梅 副教授



陈家胜 副教授



狄玉丽 副教授



张浩 副教授



石清清 副教授



荆焘 博士



代忠军 博士



明德 乐学 求实 至善

T h a n k y o u f o r l i s t e n i n g t o



西昌学院